

Numerical modelling of binary solidification by using Basilisk

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• Background

Numerical Methods

Validation tests

Research Background

• Metallurgy: binary solidification and dendritic growth (e.g. Ni-Cu alloy)



Jackson et al. 2006

Travedi 1984

• Present study

Developing accurate and sharp numerical methods
 Modelling the binary solidification

Experimental study

- Liquid metal is **opaque**
- Difficult to reveal the mechanisms controlling the solidification

Research Background —— Physical model

Temperature and concentration diffusion equations

$$\frac{\partial T_{l,s}}{\partial t} = \alpha_{l,s} \nabla^2 T_{l,s} \quad \forall \mathbf{x} \in \Omega_{l,s}$$

$$\frac{\partial Y_{l,s}}{\partial t} = D_{l,s} \nabla^2 Y_{l,s} \quad \forall \mathbf{x} \in \Omega_{l,s}$$

• Gibbs-Thomson equation

$$T_{\Gamma} = T_m + m_L Y_l |_{\Gamma} - \epsilon_{\kappa} \kappa - \epsilon_V V_{\Gamma} \quad \forall \mathbf{x} \in \Gamma$$

Stefan equation

 $\rho_s L V_{\Gamma} = (k_s \nabla T_s - k_l \nabla T_l) \cdot \mathbf{n} \quad \forall \mathbf{x} \in \Gamma$

Solute rejection relation

$$(1-k_p)Y_lV_{\Gamma} = (D_s\nabla Y_s - D_l\nabla Y_l)\cdot\mathbf{n} \quad \forall \mathbf{x} \in \Gamma$$



Problem:

How to design sharp schemes to capture the jump conditions across the interface?

Research Background — Diffused interface method



Interface is represented by the smooth function φ or T, with source terms
 No need to impose jump conditions at the solid-liquid interface

The underlying physics near the interface is **undermined**!

Research Background —— Sharp interface method







 $\frac{\partial \phi}{\partial t} \perp n \nabla \phi = 0$ *Theillard et al.* 2015 *JSC*

Level-set Method + Ghost Fluid Method





Udaykumar et al. 1999 JCP

Front-Tracking Method + Immersed Solid Method

Research Background —— Sharpness or Smoothness



Diffused interface method (one-fluid model)

 $(\partial_t + \boldsymbol{u} \cdot \nabla) T = \alpha \nabla^2 T + St^{-1} \ddot{\boldsymbol{m}} \quad \boldsymbol{x} \in \Omega.$

- Solve one governing equation in the whole domain
- The jump conditions are transferred into source terms



Sharp interface method (two-fluid model)

$$\begin{aligned} (\partial_t + \boldsymbol{u} \cdot \nabla) T^{\ell} &= \alpha^{\ell} \nabla^2 T^{\ell} \quad \boldsymbol{x} \in \Omega^{\ell} \\ \partial_t T^s &= \alpha^s \nabla^2 T^s \quad \boldsymbol{x} \in \Omega^s \\ \rho_s L V_{\Gamma} &= ((k \nabla T)^s - (k \nabla T)^{\ell}) \cdot \boldsymbol{n} \quad \boldsymbol{x} \in \Gamma \end{aligned}$$

• The jump conditions are imposed sharply at the interface

Objectives

To develop a VOF-based sharp interface method for modelling binary solidification

Advantages of the VOF-based methods:

- Conservativeness
- □ Sharp interface representation

Combined with EBM to impose sharp jump conditions







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Our recent progress of code implementation Basilisk/Gerris

Numerical methods

1. 3D Contact angle



top view of the droplet impacting

2. Evaporation flows



Impacting onto hot pool: DNS of Leidenfrost effect



Contact line hysteresis



Leidenfrost effect onto solid substrate



3. Binary solidification





Physical model

• Diffusion equations

$$\frac{\partial T_{l,s}}{\partial t} = \alpha_{l,s} \nabla^2 T_{l,s} \quad \mathbf{x} \in \Omega_{l,s}, \quad (1)$$

$$\frac{\partial Y_{l,s}}{\partial t} = D_{l,s} \nabla^2 Y_{l,s} \quad \mathbf{x} \in \Omega_{l,s}. \quad (2)$$

• Jump conditions

$$L_{H}V_{\Gamma} = (k_{s}\nabla T_{s} - k_{l}\nabla T_{l}) \cdot \mathbf{n}, \quad (3)$$

(1 - k_p) $Y_{l,\Gamma}V_{\Gamma} = -(D_{l}\nabla Y_{l}) \cdot \mathbf{n}. \quad (4)$

• Gibbs-Thomson equation

$$T_{\Gamma} = \underbrace{T_{m}}_{\text{Melting}} + \underbrace{m_{L}Y_{\Gamma}}_{\text{Solute}} + \underbrace{\epsilon_{c}\kappa}_{\text{Capillary}} + \underbrace{\epsilon_{v}V_{\Gamma}}_{\text{Kinetic}} .$$
 (5)
Melting Temp. Conc. Undercooling Undercooling



Difficulty: Sharp jump conditions Coupling of T and Y at interface

Sharp interface method



Diffused interface method (one-fluid model)

 $(\partial_t + \boldsymbol{u} \cdot \nabla) T = \alpha \nabla^2 T + St^{-1} \ddot{\boldsymbol{m}} \quad \boldsymbol{x} \in \Omega.$

- Solve one governing equation in the whole domain
- The jump conditions are transferred into source terms



Sharp interface method (two-fluid model)

 $(\dot{\epsilon}$

$$\partial_t + \boldsymbol{u} \cdot \nabla T^{\ell} = \alpha^{\ell} \nabla^2 T^{\ell} \quad \boldsymbol{x} \in \Omega^{\ell}$$

 $\partial_t T^s = \alpha^s \nabla^2 T^s \quad \boldsymbol{x} \in \Omega^s$
 $\rho_s LV_{\Gamma} = ((k \nabla T)^s - (k \nabla T)^{\ell}) \cdot \boldsymbol{n} \quad \boldsymbol{x} \in \Omega$

• The jump conditions are imposed sharply at the interface

Global procedures

For binary fluid, the diffusivity of solute is much smaller than that of temperature. To avoid numerical oscillation, the front velocity need to be computed according to the solute.

Velocity >> Concentration >> Temperature >> Velocity	Velocity >> Temperature >> Concentration >> Velocity
1. $Y^{n+1} \leftarrow V_{\Gamma}^{n}$ (Robin BC)	1. $T^{n+1} \leftarrow V_{\Gamma}^{n}$ (Jump condition)
$(1 - k_p) Y_I V_{\Gamma} = -D_I \bigvee Y_I \cdot \mathbf{n}.$ 2. $T^{n+1} \leftarrow Y^{n+1}$ (Dirichlet BC)	$L_H V_{\Gamma} = (\kappa_s \vee I_s - \kappa_l \vee I_l) \cdot \mathbf{n}.$ 2. $Y^{n+1} \leftarrow T^{n+1}$ (Dirichlet BC)
$T_{\Gamma} = T_m + m_L Y_{\Gamma} + \epsilon_c(\theta) \kappa + \epsilon_v V_{\Gamma}.$	$T_{\Gamma} = T_m + m_L Y_{\Gamma} + \epsilon_c(\theta) \kappa + \epsilon_v V_{\Gamma}.$
3. $V^{n+1} \leftarrow T^{n+1}$ (Interpolation) $V_{\Gamma} = \frac{1}{L_H} \left(k_s \nabla T_s - k_l \nabla T_l \right) \cdot \mathbf{n}.$	3. $V^{n+1} \leftarrow Y^{n+1}$ (Interpolation) $V_{\Gamma} = -\frac{D_l}{(1-k_p)Y_l} \nabla Y_l \cdot \mathbf{n}.$

Numerical method – Flux jump conditions

Given jump conditions from Stefan condition

$$[\vartheta \nabla \chi]_{\Gamma} = \left((\vartheta \nabla_{\Gamma} \chi)_{s} - (\vartheta \nabla_{\Gamma} \chi)_{l} \right) \cdot \boldsymbol{n} = \gamma \quad \forall \boldsymbol{x} \in \Gamma$$
$$[\chi]_{\Gamma} = 0 \quad \forall \boldsymbol{x} \in \Gamma$$

• Finite volume method + Divergence theorem



Relation between cell centered values for two phases

$$(\chi_l)_i - (\chi_s)_i = \frac{(\Delta + \xi_{i+1}) \left(\frac{\delta \gamma}{\vartheta}\right)_i - \xi_i \left(\frac{\delta \gamma}{\vartheta}\right)_{i+1}}{\xi_i - \xi_{i+1} - \Delta} + O(\Delta^2)$$







Background

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Numerical tests – 2D unstable solidification

• Anisotropic capillary coefficient

 $\epsilon_{\kappa} = 0.5(1 - 0.75\cos 4\theta)$

Predicted tip velocity

 $V_{tip} = \begin{cases} 0.034, & T_l = -0.55\\ 0.0935 & T_l = -0.65 \end{cases}$





Numerical tests – 2D unstable binary solidification

(b)

-0.55

-0.45 -0.4 -0.35 -0.3 -0.25 -0.18

(d)

0.26

-(a)

works

Adaptive mesh (c) Solutal BL is Comparison with other much thinner! Present model Ramirez and Beckermann 2005 -----0.15 0.2 0.1 Tan and Zabaras 2007

Temperature field

Concentration field

Numerical tests – 3D unstable binary solidification





Planar growth (stable)



Cellular growth (unstable)



Dendritic growth (unstable)



Numerical tests – Melting ice sphere under forced convection



Black solid line: Initial interface

Red Solid line: Present work

Unfilled points: Experimental work

Melting in 2.5°C water
 Melting in 2.5°C NaCl solution



Y. L. Hao, Y.-X. Tao, Journal of heat transfer, 2001

Numerical tests – Melting ice sphere under forced convection



Conclusion

- VOF-based sharp method for modelling binary solidification;
- □ Sharp flux jump of temperature at the solid-liquid interface;
- Unstable binary dendritic growth with high Lewis number;
- Powerful numerical tools in metallurgic applications

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Thanks for your attention

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COLUMN TWO IS NOT THE OWNER.